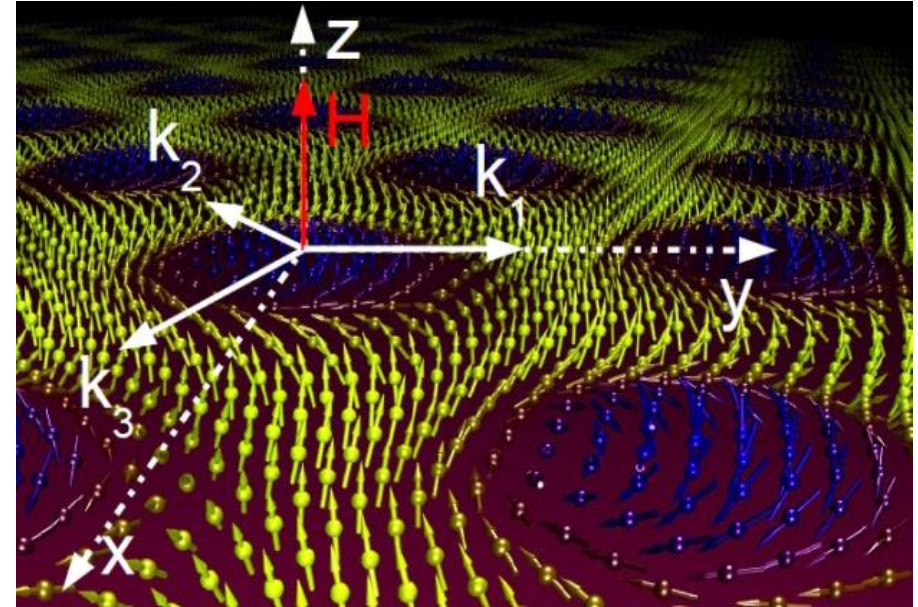
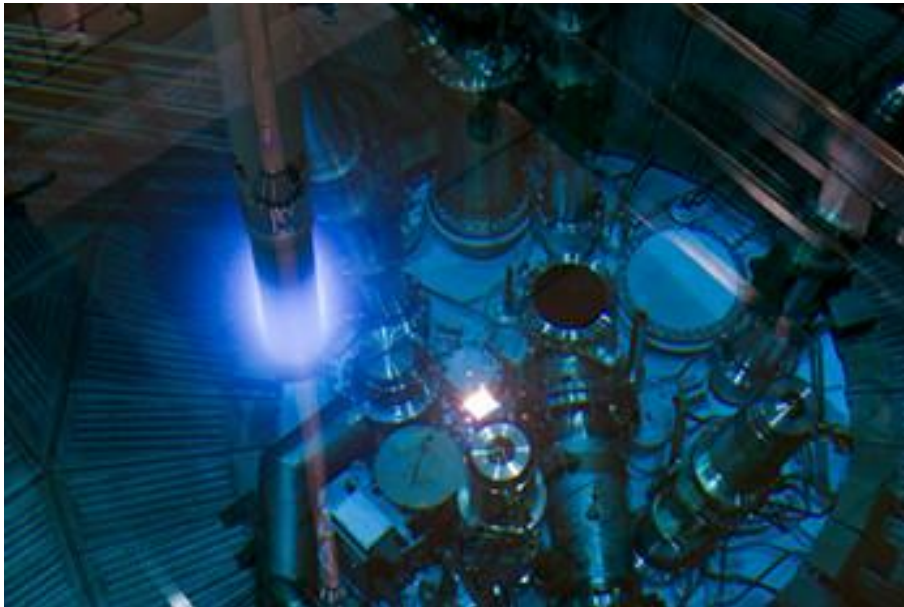




# Physics with Neutrons I, WS 2015/2016



## Lecture 7, 30.11.2015

MLZ is a cooperation between:



## Visit FRM II : 21.12.2015

10:15 – 13:00 (after the lecture)

Valid ID necessary!!!

## Exam (after winter term)

➡ Registration: via TUM-Online between 16.11.2015 – 15.1.2015

➡ Email: [sebastian.muehlbauer@frm2.tum.de](mailto:sebastian.muehlbauer@frm2.tum.de) for date arrangement

➡ 30min oral exam

#### 3.3 Reminder: Coherent and incoherent scattering

Assume a large sample with statistically variations of  $b_j$

➔ No correlations among  $b$ :  $\overline{b_{j'} b_j} = (\overline{b})^2, j' \neq j$

$$\overline{b_j b_j} = (\overline{b^2}), j' = j$$

Double differential crosssection splits up into two terms:

$$\left( \frac{d^2\sigma}{d\Omega dE'} \right)_{coherent} = \frac{\sigma_{coh.}}{4\pi} \frac{k'}{k} \frac{1}{2\pi\hbar} \sum_{j,j'} \int_{-\infty}^{\infty} \langle \underline{e^{i\kappa R_{j'}(0)} e^{-i\kappa R_j(t)}} \rangle e^{-i\omega t} dt$$

$$\left( \frac{d^2\sigma}{d\Omega dE'} \right)_{incoherent} = \frac{\sigma_{inc.}}{4\pi} \frac{k'}{k} \frac{1}{2\pi\hbar} \sum_{j,j'} \int_{-\infty}^{\infty} \langle \underline{e^{i\kappa R_j(0)} e^{-i\kappa R_j(t)}} \rangle e^{-i\omega t} dt$$

$$\sigma_{coh.} = 4\pi \overline{b}^2 \quad \sigma_{inc.} = 4\pi (\overline{b^2} - \overline{b}^2)$$

#### 3.3 Reminder: Coherent and incoherent scattering

##### Coherent



Spatial and temporal correlations between different atoms

- ➡ Interference effects:
- ➡ Given by average of  $b$  Bragg scattering

##### Incoherent



Spatial and temporal correlations between the same atom

- ➡ Constant in  $Q$
- ➡ Given by variations in  $b$  due to spin, disorder, random atomic motion....

## Reminder: 4.1 Elastic scattering from a single crystal

➔ Final result for elastic coherent scattering on crystals

$$\frac{d\sigma}{d\Omega_{coh.el}} = N_0 \frac{(2\pi)^3}{v_0} e^{-2W(\vec{\kappa})} \sum_{\vec{\tau}} |S_{\vec{\tau}}|^2 \delta(\vec{\kappa} - \vec{\tau})$$

Normalization

Debye-Waller

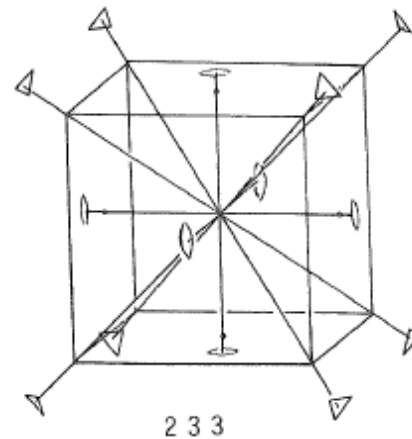
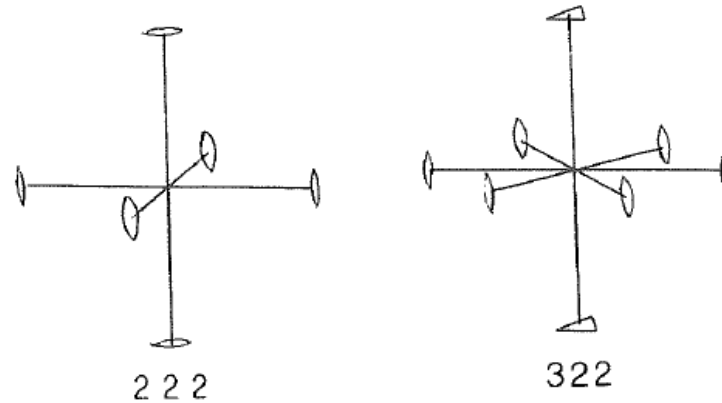
Structure factor

Bragg positions



## 4.2 Appetizer Crystallography

Euler rules: Combination of rotations



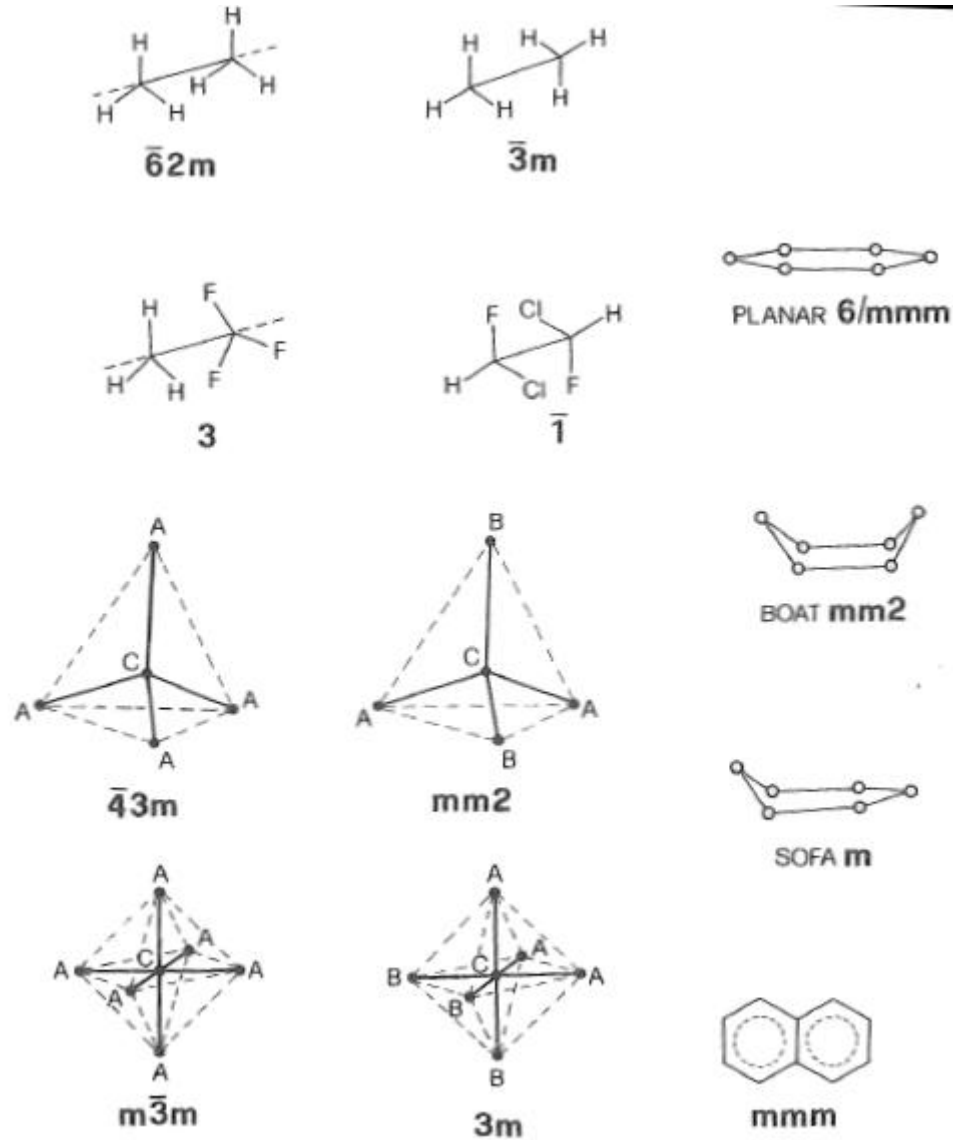
#### 4.2 Appetizer Crystallography: 32 Point groups

Crystal systems	Point groups		Laue classes	Lattice point groups
	Non-centro-symmetric	Centro-symmetric		
Triclinic	1	$\bar{1}$	$\bar{1}$	$\bar{1}$
Monoclinic	2	m	2/m	2/m
Orthorhombic	222	mm2	mmm	mmm
Tetragonal	4	$\bar{4}$	4/m	4/mmm
	422	4mm, $\bar{4}2m$	4/mmm	
Trigonal	3	$\bar{3}$	$\bar{3}$	$\bar{3}m$
	32	3m	$\bar{3}m$	
Hexagonal	6	$\bar{6}$	6/m	6/mmm
	622	6mm, $\bar{6}2m$	6/mmm	
Cubic	23	$\bar{m}\bar{3}$	$\bar{m}\bar{3}$	$\bar{m}\bar{3}m$
	432	$\bar{4}3m$	$\bar{m}\bar{3}m$	



## 4.2 Appetizer Crystallography

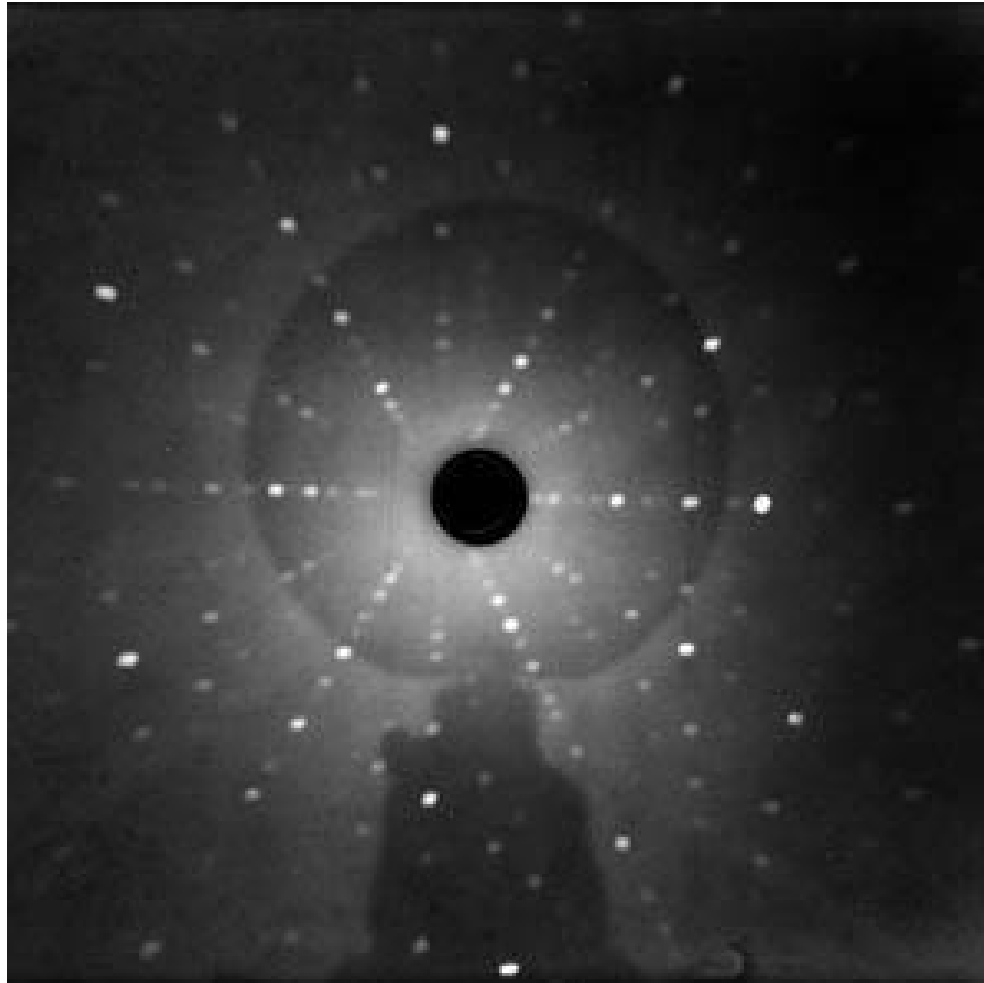
Point groups: examples



#### 4.2 Appetizer Crystallography: Point groups & Laue classes

Crystal systems	Point groups		Laue classes	Lattice point groups
	Non-centro-symmetric	Centro-symmetric		
Triclinic	1	$\bar{1}$	$\bar{1}$	$\bar{1}$
Monoclinic	2	m	2/m	2/m
Orthorhombic	222	mm2	mmm	mmm
Tetragonal	4	$\bar{4}$	4/m	4/mmm
	422	4mm, $\bar{4}2m$	4/mmm	
Trigonal	3		$\bar{3}$	$\bar{3}m$
	32	3m	$\bar{3}m$	
Hexagonal	6	$\bar{6}$	6/m	6/mmm
	622	6mm, $\bar{6}2m$	6/mmm	
Cubic	23		$m\bar{3}$	$m\bar{3}m$
	432	$\bar{4}3m$	$m\bar{3}m$	

#### 4.2 Appetizer Crystallography: Laue classes, Friedels law



Laue forward scattering

Silicon single crystal, cubic, fcc,  
[111], three fold

#### 4.2 Appetizer Crystallography: 7 crystal classes

**Table 1.8.** The conventional types of unit cell

Symbol	Type	Positions of additional lattice points	Number of lattice points per cell
P	primitive	—	1
I	body centred	$(1/2, 1/2, 1/2)$	2
A	A-face centred	$(0, 1/2, 1/2)$	2
B	B-face centred	$(1/2, 0, 1/2)$	2
C	C-face centred	$(1/2, 1/2, 0)$	2
F	All faces centred	$(1/2, 1/2, 0), (1/2, 0, 1/2), (0, 1/2, 1/2)$	4
R	Rhombohedrally centred (description with 'hexagonal axes')	$(1/3, 2/3, 2/3), (2/3, 1/3, 1/3)$	3

## 4.2 Appetizer Crystallography: 14 Bravais Lattices

The 7 lattice systems	The 14 Bravais lattices			
triclinic	P			
	$\alpha, \beta, \gamma \neq 90^\circ$			
monoclinic	P	C		
	$\alpha \neq 90^\circ, \beta, \gamma = 90^\circ$	$\alpha \neq 90^\circ, \beta, \gamma = 90^\circ$		
orthorhombic	P	C	I	F
	$a \neq b \neq c$	$a \neq b \neq c$	$a \neq b \neq c$	$a \neq b \neq c$
	$a$ $b$ $c$	$a$ $b$ $c$	$a$ $b$ $c$	$a$ $b$ $c$

tetragonal	P	I		
	$a \neq c$	$a \neq c$		
rhombohedral	P			
	$\alpha = \beta = \gamma \neq 90^\circ$			
hexagonal	P			
cubic	P (bcc)	I (bcc)	F (fcc)	
	$a$	$a$	$a$	

## 4.2 Appetizer Crystallography: 320 space groups

Table 1.9. The 230 three-dimensional space groups arranged by crystal systems and point groups. Space groups (and enantiomorphous pairs) that are uniquely determinable from the symmetry of the diffraction pattern and from systematic absences (see p. 159) are shown in bold-type. Point groups without inversion centres or mirror planes are emphasized by boxes

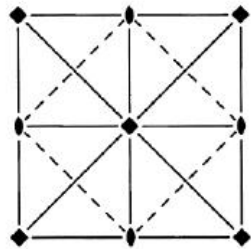
Crystal system	Point group	Space groups
Triclinic	<b>1</b>	P1
	$\bar{1}$	P $\bar{1}$
Monoclinic	<b>2</b>	P2, P2 <sub>1</sub> , C2
	m	Pm, Pc, Cm, Cc
	2/m	P2/m, P2 <sub>1</sub> /m, C2/m, P2/c, P2 <sub>1</sub> /c, C2/c
Orthorhombic	<b>222</b>	P222, P222 <sub>1</sub> , P2 <sub>1</sub> 2 <sub>1</sub> 2, P2 <sub>1</sub> 2 <sub>1</sub> 2 <sub>1</sub> , C222 <sub>1</sub> , C222, F222, I222, I2 <sub>1</sub> 2 <sub>1</sub> 2 <sub>1</sub>
	mm2	Pmm2, Pmc2 <sub>1</sub> , Pcc2, Pma2 <sub>1</sub> , Pca2 <sub>1</sub> , Pnc2 <sub>1</sub> , Pmn2 <sub>1</sub> , Pba2, Pna2 <sub>1</sub> , Pnn2, Cmm2, Cmc2 <sub>1</sub> , Ccc2, Amm2, Abm2, Ama2, Aba2, Fmm2, Fdd2, Imm2, Iba2, Ima2
	mmm	Pmmm, Pnnn, Pccm, Pban, Pmma, Pnna, Pmna, Pcca, Pbam, Pccn, Pbcm, Pnmm, Pmma, Pbcn, Pbca, Pnma, Cmcm, Cmca, Cmmm, Cccm, Cmma, Ceca, Fmmm, Fddd, Immm, Ibam, Ibea, Imma
Tetragonal	<b>4</b>	P4, P4 <sub>1</sub> , P4 <sub>2</sub> , P4 <sub>3</sub> , I4, I4 <sub>1</sub>
	4	P4, I4
	4/m	P4/m, P4 <sub>2</sub> /m, P4/n, P4 <sub>2</sub> /n, I4/m, I4 <sub>1</sub> /a
	<b>422</b>	P422, P4 <sub>2</sub> 2, P4 <sub>1</sub> 22, P4 <sub>2</sub> 2 <sub>1</sub> 2, P4 <sub>2</sub> 22, P4 <sub>2</sub> 2 <sub>1</sub> 2, P4 <sub>3</sub> 22, P4 <sub>3</sub> 2 <sub>1</sub> 2, I422, I4 <sub>1</sub> 22
	4mm	P4mm, P4bm, P4 <sub>2</sub> cm, P4 <sub>2</sub> nm, P4cc, P4nc, P4 <sub>2</sub> mc, P4 <sub>2</sub> bc, I4mm, I4cm, I4 <sub>1</sub> md, I4 <sub>1</sub> ed
	4m	P42m, P4 <sub>2</sub> c, P4 <sub>2</sub> m, P4 <sub>2</sub> e, P4m2, P4c2, P4b2, P4n2, I4m2, I4c2, I42m, I42d
4/mmm	P4/mmm, P4/mcc, P4/nbm, P4/nnc, P4/mbm, P4/mnc, P4/nmm, P4/ncc, P4 <sub>2</sub> /mmc, P4 <sub>2</sub> /mcm, P4 <sub>2</sub> /nbc, P4 <sub>2</sub> /nmm, P4 <sub>2</sub> /mbc, P4 <sub>2</sub> mnm, P4 <sub>2</sub> /nmc, P4 <sub>2</sub> /ncm, I4/mmm, I4/mcm, I4 <sub>1</sub> /amd, I4 <sub>1</sub> /acd	
Trigonal-hexagonal	<b>3</b>	P3, P3 <sub>1</sub> , P3 <sub>2</sub> , R3
	3	P3, R3
	<b>32</b>	P312, P321, P3 <sub>1</sub> 12, P3 <sub>2</sub> 21, P3 <sub>2</sub> 12, P3 <sub>2</sub> 21, R32
	3m	P3m1, P31m, P3c1, P31c, R3m, R3c
	3m	P31m, P31c, P3m1, P3c1, R3m, R3c
	<b>6</b>	P6, P6 <sub>1</sub> , P6 <sub>5</sub> , P6 <sub>3</sub> , P6 <sub>2</sub> , P6 <sub>4</sub>
	6	P6
	6/m	P6/m, P6 <sub>3</sub> /m
	<b>622</b>	P622, P6 <sub>1</sub> 22, P6 <sub>5</sub> 22, P6 <sub>3</sub> 22, P6 <sub>2</sub> 22, P6 <sub>4</sub> 22
	6mm	P6mm, P6cc, P6 <sub>3</sub> cm, P6 <sub>3</sub> mc
6m	P6m2, P6c2, P62m, P62c	
6/mmm	P6/mmm, P6/mcc, P6 <sub>3</sub> /mcm, P6 <sub>3</sub> /mmc	
Cubic	<b>23</b>	P23, F23, I23, P2 <sub>3</sub> , I2 <sub>3</sub>
	$m\bar{3}$	Pm3, Pn3, Fm3, Fd3, Im3, Pa3, Ia3
	<b>432</b>	P432, P4 <sub>3</sub> 2, F432, F4 <sub>3</sub> 2, I432, P4 <sub>3</sub> 32, P4 <sub>3</sub> 2, I4 <sub>3</sub> 2
	43m	P43m, F43m, I43m, P43n, F43c, I43d
	$m\bar{3}m$	Pm3m, Pn3n, Pm3n, Pn3m, Fm3m, Fm3c, Fd3m, Fd3c, Im3m, Ia3d

## 4.2 Appetizer Crystallography:

### International tables of crystallography

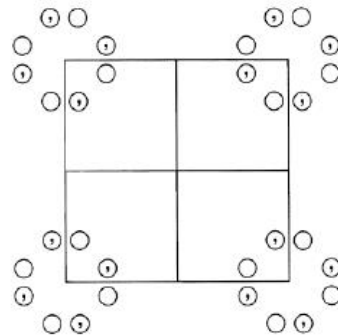
$p4mm$

No. 11



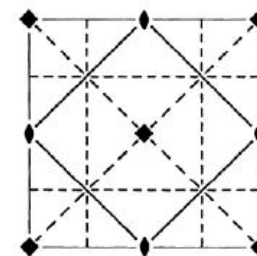
$4mm$

$p4mm$



Patterson

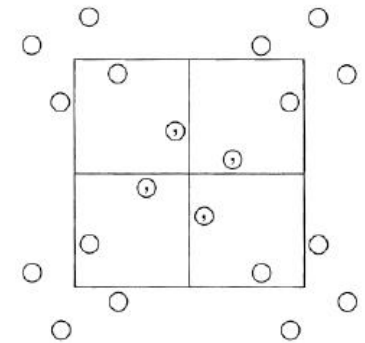
Patterson symmetry  $p4mm$



$4mm$

$p4gm$

$p4gm$



Origin at  $4mm$

Asymmetric unit  $0 \leq x \leq \frac{1}{2}; 0 \leq y \leq \frac{1}{2}; x \leq y$

Symmetry operations

- |                  |                        |                  |                        |
|------------------|------------------------|------------------|------------------------|
| (1) $x, y$       | (2) $\bar{x}, \bar{y}$ | (3) $\bar{y}, x$ | (4) $y, \bar{x}$       |
| (5) $\bar{x}, y$ | (6) $x, \bar{y}$       | (7) $y, x$       | (8) $\bar{y}, \bar{x}$ |

Origin at  $4lg$

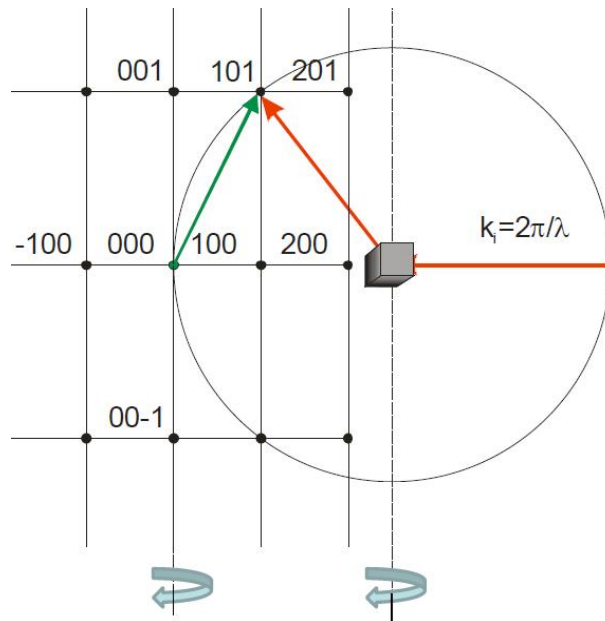
Asymmetric unit  $0 \leq x \leq \frac{1}{2}; 0 \leq y \leq \frac{1}{2}; y \leq \frac{1}{2} - x$

Symmetry operations

- |                                              |                                              |                                        |                                                    |
|----------------------------------------------|----------------------------------------------|----------------------------------------|----------------------------------------------------|
| (1) $x, y$                                   | (2) $\bar{x}, \bar{y}$                       | (3) $\bar{y}, x$                       | (4) $y, \bar{x}$                                   |
| (5) $\bar{x} + \frac{1}{2}, y + \frac{1}{2}$ | (6) $x + \frac{1}{2}, \bar{y} + \frac{1}{2}$ | (7) $y + \frac{1}{2}, x + \frac{1}{2}$ | (8) $\bar{y} + \frac{1}{2}, \bar{x} + \frac{1}{2}$ |

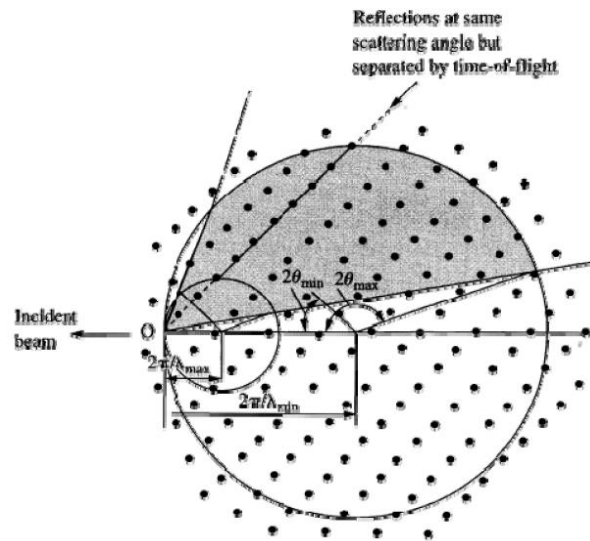
#### 4.3 Diffraction: Monochromatic vs. TOF vs. Laue

##### Monochromatic beam



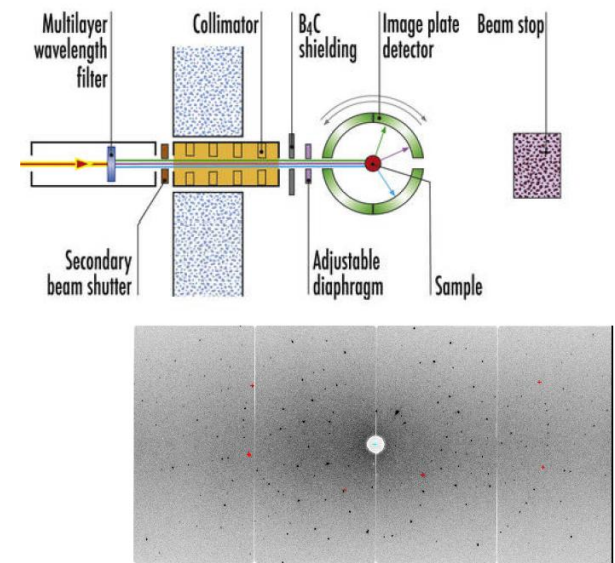
- ➡ Ewald construction
- ➡ Less intensity
- ➡ Rocking curve gives intensity of Bragg peak
- ➡ Clean data

##### Time-of-flight (TOF)



- ➡ Ewald construction for each wavelength in the beam
- ➡ Rocking curve distributed in time and detector
- ➡ Waste less neutrons

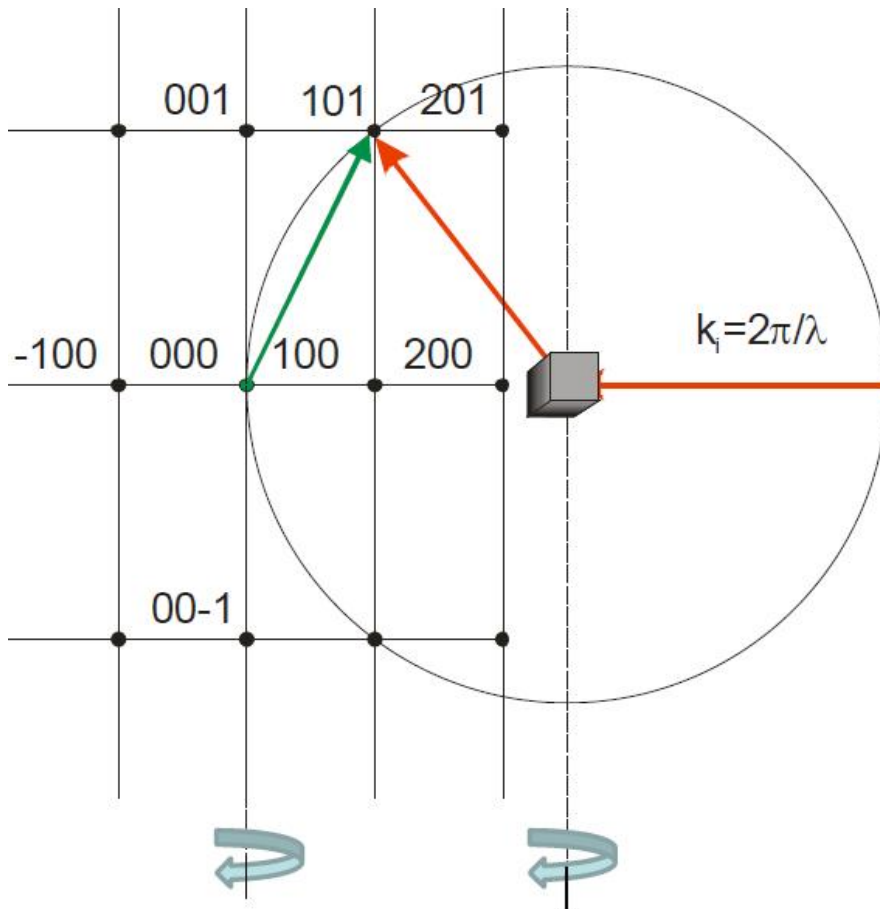
##### Laue (polychromatic beam)



- ➡ Essentially white beam
- ➡ More Bragg peaks (not stronger)
- ➡ Hard to get intensities
- ➡ Large background



#### 4.3 Diffraction: Ewald construction for single crystal diffraction

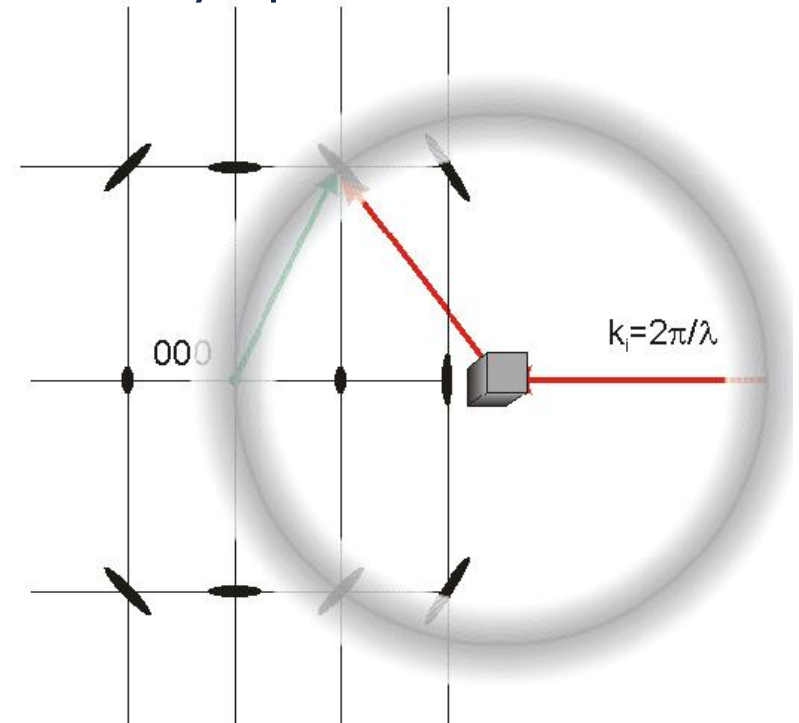
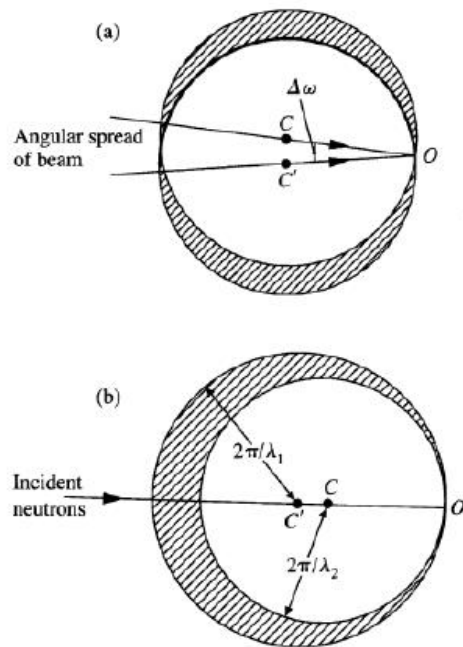


Ewald construction:

- ➔ Sphere around center of crystal (in real space)
- ➔ Origin of reciprocal space on transmitted beam at the edge of Ewald sphere
- ➔ Rotation of crystal (real space) corresponds to a rotation of reciprocal space

#### 4.3 Diffraction: Ewald construction including resolution effects

- ➔ Mosaic spread of the sample (no perfect crystal)
- ➔ Finite spread of incoming beam (wavelength spread)
- ➔ Finite collimation of the incoming beam
- ↳ Ewald construction for real samples: Fuzzy sphere



#### 4.3 Diffraction: Crystal rotation method

➔ Solution: Measurement of **integrated intensity** and not **peak intensity**

➔ Fuzziness integrated out

➔ Rocking scan

